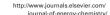


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# Morphology and crystal-plane effects of Zr-doped CeO<sub>2</sub> nanocrystals on the epoxidation of styrene with tert-butylhydroperoxide as the oxidant

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#### ABSTRACT

The morphology effect of Zr-doped  $CeO_2$  was studied in terms of their activities in the selective oxidation of styrene to styrene oxide using tert-butyl hydroperoxide as the oxidant. In the present work, Zr-doped  $CeO_2$  nanorods exhibited the highest catalytic performance (yield of styrene oxide and TOF value) followed by nanoparticles and nanocubes. For the Zr-doped  $CeO_2$  nanorods, the apparent activation energy is 56.3 kJ/mol, which is much lower than the values of catalysts supported on nanoparticles and nanocubes (73.3 and 93.4 kJ/mol). The high resolution transmission electron microscopy results indicated that (100) and (110) crystal planes are predominantly exposed for Zr-doped  $CeO_2$  nanorods while (100) and (111) for nanocubes, (111) for nanoparticles. The remarkably increased catalytic activity of the Zr-doped  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is mainly attributed to the higher percentage of  $CeO_2$  nanorods is initial high catalytic activity after five successive recycles.

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# 1. Introduction

Epoxidation reaction of styrene plays an important role in industrial processes because the products from epoxidation reaction of styrene are versatile intermediates in fine chemicals for the synthesis of perfumes, plasticizers, pharmaceuticals and drugs, etc. [1]. Traditionally, the epoxidation of styrene is carried out by the direct oxidation of alkenes with stoichiometric peracids which produced much waste and byproducts, and even worse, the peracids are corrosive, hazardous to the environment. With growing attention paid to the sustainable development, much more efforts have been made to develop a variety of new processes of styrene epoxidation using H<sub>2</sub>O<sub>2</sub>, urea-H<sub>2</sub>O<sub>2</sub> or TBHP as oxidants with the catalysts such as TS-1 [8], Ti-SiO<sub>2</sub> [9], Ti-HMS [10], as well as  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> [11]. Meanwhile, the separable solid catalysts, such as Mo [2], Fe [3], Mn [4], Co [5], Cu [6] and Ni [7] based catalysts, were proven to be efficient in the epoxidation of styrene. However, the catalysts mentioned above exhibited comparatively unsatisfied epoxide selectivity or low activity [8–11].

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As a well-known functional rare earth material, ceria has been widely applied in the fields of industrial heterogeneous catalysts, such as automotive three-way exhaust (TWCs) [12], water-gas-shift (WGS) reactions [13], solid oxide fuel cells anodes (SOFCs) [14]. In order to promote the physical and chemical properties of CeO<sub>2</sub>, incorporation of cheaper transition metals such as Mn, Fe, Co, Ni, and Zr by doping approach is a suitable option because of the low cost and strong stability of transition metals against poisoning. For example, recently, Chen et al. [15] reported the shape effect of Ce1-xZrxO2 nanorods in terms of redox features and catalytic activity in CO oxidation. Besides, other groups reported that CeO2-CuO nanomaterials showed high catalytic activity when compared with pure CeO<sub>2</sub> [16]. The insertion of Zr into ceria lattice leads to structural distortions of the cubic fluorite lattice compared with pure CeO<sub>2</sub>, ultimately resulting in the enhancement of mobility of lattice oxygen and the density of oxygen vacancies. CeO2 in a cubic fluorite structure possesses three low-index planes: (110), (111) and (100); the (111) plane is the most stable, followed by (110) and (100) facets [17-19]. As we know, ceria catalysts with different morphologies were established to possess different crystal planes and surface properties [20], which can influence the interaction between ceria surface and the adsorbed reactants. For instance, Zhou et al. found that CeO<sub>2</sub> nanorods showed better

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catalytic performance in CO oxidation than irregular nanoparticles because of the exposed reactive crystal planes [17]. Moraes et al. reported the effects of ceria morphology on catalytic performance of Ni/CeO<sub>2</sub> for the low temperature steam reforming of ethanol and found that Ni supported on ceria nanocubes had the best performance [21]. Fan et al. found that spindle-like CeO<sub>2</sub> showed a higher catalytic activity on CO conversion [22]. So far, the morphology dependence on Zr-doped CeO<sub>2</sub> nanomaterials for the epoxidation reaction of styrene has been rarely reported yet.

In our previous work, the CeO<sub>2</sub> nanorods with various Zr dopant concentrations catalysts are prepared and investigated in terms of their catalytic activities in catalyzing styrene oxidation [23]. In the present work, Zr-doped CeO<sub>2</sub> catalysts of different morphologies are further investigated in terms of their activities in catalyzing the styrene oxidation reaction. The structure of the Zr-doped CeO<sub>2</sub> is studied using various techniques including Brunauer-Emmett-Teller (BET) method, X-ray diffraction (XRD), transmission electron microscopy (TEM), high-resolution transmission electronmicroscopy (HRTEM), Raman spectroscopy, and X-ray photoelectron spectroscopy (XPS) in order to investigate the relationship between the morphology and the catalytic activity. It is found that the morphology has a great influence on the exposed crystal planes, oxygen vacancy concentrations and the content of Ce<sup>3+</sup> ions, which contribute to the variation of the catalytic activity in styrene oxidation reaction.

#### 2. Experimental

#### 2.1. Catalyst preparation

All the materials were purchased from Sinopharm Chemical Reagent Co., Ltd. China. All chemicals were used without further purification. Deionized water was used throughout the experiments.

Zr-doped CeO<sub>2</sub> nanorods, Zr-doped CeO<sub>2</sub> nanocubes, and Zrdoped CeO<sub>2</sub> nanoparticles were prepared by a hydrothermal method varying the amount of reactant and hydrothermal temperature. The theoretical molar ratios of Zr/(Zr+Ce) over the Zr-doped CeO<sub>2</sub> were all about 10%. Typically, 4.5 mmol of Ce(NO<sub>3</sub>)<sub>3</sub>•6H<sub>2</sub>O, 0.5 mmol of Zr(NO<sub>3</sub>)<sub>4</sub>•5H<sub>2</sub>O, and an appropriate amount of NaOH (19.2 g for nanorods and nanocubes, 0.32 g for nanoparticles) were dissolved into 40 mL of distilled water, respectively. Then, the solution was mixed together and kept stirring for 30 min. The mixture was then transferred to a stainless steel reaction tank with polytetrafluoroethylene (PTFE) lining (100 mL) and maintained at an appropriate temperature (373 K for nanorods, 453 K for nanocubes and nanoparticles) for 24 h. After centrifugation and washing with water and ethanol, the precipitate was dried at 353 K overnight, and finally calcined at 673 K for 5 h in air with the ramping rate of 2 °C/min. The as-prepared Zr-doped CeO<sub>2</sub> nanorods, Zr-doped CeO<sub>2</sub> nanoparticles and Zr-doped CeO2 nanocubes were respectively denoted as Zr-doped CeO<sub>2</sub> NR, Zr-doped CeO<sub>2</sub> NP and Zr-doped CeO<sub>2</sub>

# 2.2. Characterizations

XRD patterns were obtained on a Bruker D8 advance spectrometer with Cu  $K\alpha$  radiation ( $\lambda = 0.154$  nm), operated at 40 mA and 40 kV, respectively. The lattice parameter a value was calculated using the Scherrer equation. TEM images were performed on a JOEL JEM 2010 transmission electron microscope. The samples were supported on carbon-coated copper grids for the experiment. XPS was performed using a RBD 147 upgraded Perkin Elmer PHI 5000C ESCA system equipped with a dual X-ray source, of which the Mg  $K\alpha$  (1253.6 eV) anode and a hemispherical energy analyzer was used. The background pressure during data acquisition was

maintained at  $<10^{-6}$  Pa. Measurements were performed at a pass energy of 93.90 eV. All binding energies were calibrated using contaminant carbon (C 1s=284.6 eV). The actual composition of the catalysts was determined by Optima 8000 ICP-OES Spectrometer. The laser Raman experiments were obtained using a Jobin Yvon Dilor Labram I Raman spectrometer equipped with a charge coupled device detector, holographic notch filter, and the He-Ne laser radiating at 632.8 nm. Specific surface areas of the catalysts were measured by nitrogen adsorption-desorption at 77 K (Micromeritics Tristar ASAP 3000) employing the BET method. The sample was pre-treated at 523 K in vacuum for 3 h before test to degas the adsorbed moisture.

#### 2.3. Activity test

The epoxidation of styrene was carried out at reflux oil bath temperature of 353 K in a closed 25 mL regular glass reactor with magnetically stirred (400 r/min) using 70 wt% aqueous *t*-butyl hydroperoxide (TBHP) as the oxidant and acetonitrile as solvent. First, 0.05 g of the catalyst, 8.7 mmol of 70 wt% aqueous TBHP and 10 mL of acetonitrile were introduced into the regular glass reactor at 353 K with magnetic stirring. Then, 4.35 mmol of styrene was added into the above mixture to start the reaction and the mixture was maintained for a period of time. Gas chromatography (GC) was used to carry out the quantitative analysis of the reaction products, and GC-mass spectroscopy (GC-MS) was applied to determine different products in the reaction mixture.

Substrate conversion = moles of substrate converted/moles of substrate used;

Product selectivity = moles of product formed/moles of substrate converted;

The stability was tested as follows: the catalyst was separated by centrifuge, washed by water and ethanol for several times, and dried under 353 K for 12 h.

#### 3. Results and discussion

### 3.1. Characterization of the as-prepared material

The morphology of the synthesized Zr-doped CeO<sub>2</sub> nanoparticles, Zr-doped CeO2 nanorods, and Zr-doped CeO2 nanocubes are investigated by TEM and HRTEM as shown in Fig 1. As presented in Fig 1(a), Zr-doped CeO<sub>2</sub> nanoparticles are uniformly distributed and established to be around 10 nm in diameter. Fig 1(b) depicts a high revolution TEM image of Zr-doped CeO2 nanoparticles with a fast Fourier transform analysis (inset). According to the HRTEM image, the dominating (111) planes are observed with an interplanar spacing of 0.32 nm, being accorded well with the previous report [24]. For the Zr-doped CeO<sub>2</sub> nanorods, the measured size is approximate with 30-80 nm in length and  $10 \pm 1$  nm in diameter as shown in Fig 1(c). Meanwhile, (100) and (110) crystal planes are predominantly exposed with an interplanar spacing of 0.26 and 0.19 nm according to its HRTEM image. For the Zr-doped CeO<sub>2</sub> nanocubes exhibited in Fig. 1(e) and (f), the diameter distribution is about 60-80 nm. At the same time, a small number of (100) planes and the dominating (111) planes are observed with an interplanar spacing of 0.27 and 0.32 nm. Some light points on the ceria are observed from the HRTEM image in Fig. 1(b), (d), (f), revealing the existence of the surface defects. Theoretical studies have shown that (1 1 1) is the least active surface followed by (100) and (110), in which the energy needed to produce oxygen vacancy is inverse proportional to the order of the reactivity of crystal planes [25-27]. Thus, according to the exposed crystal planes of Zr-doped CeO2 of different morphologies observed in HRTEM image, the Zr-doped CeO2 nanorods exhibited the highest concentration of surface oxygen vacancies, followed by Zr-doped

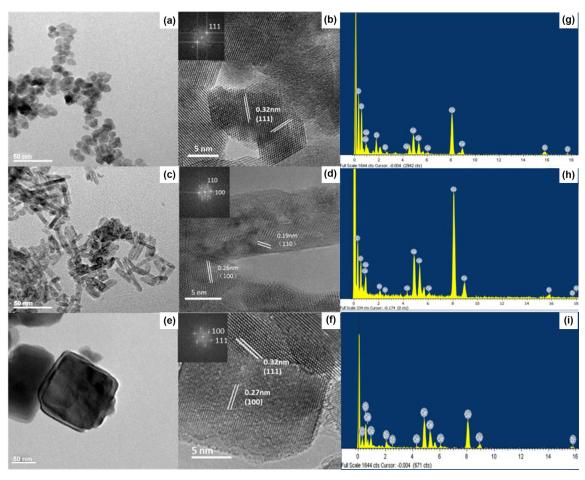


Fig. 1. TEM images of Zr-doped CeO<sub>2</sub> (a) nanoparticles, (c) nanorods, and (e) nanocubes. HRTEM image of Zr-doped CeO<sub>2</sub> (b) nanoparticles, (d) nanorods, and (f) nanocubes. EDS patterns of Zr-doped CeO<sub>2</sub> (g) nanoparticles, (h) nanorods, and (i) nanocubes.

 $CeO_2$  nanocubes and nanoparticles. The EDS spectra in Fig. 1(g), (h), (i) confirm the presence of elements Ce, Zr and O (the Cu element is from the substrates), which indicated the homogeneity of the obtained catalysts.

The crystal structures of the as-synthesized Zr-doped  $CeO_2$  nanorods, Zr-doped  $CeO_2$  nanoparticles, and Zr-doped  $CeO_2$  nanocubes are investigated by XRD techniques and shown in Fig. 2. The diffraction peaks at  $28.5^{\circ}$ ,  $33.0^{\circ}$ ,  $47.5^{\circ}$ ,  $56.3^{\circ}$  and  $59.1^{\circ}$  assigned to the  $(1\ 1\ 1)$ ,  $(2\ 0\ 0)$ ,  $(2\ 2\ 0)$ ,  $(3\ 1\ 1)$  and  $(2\ 2\ 2)$  planes are

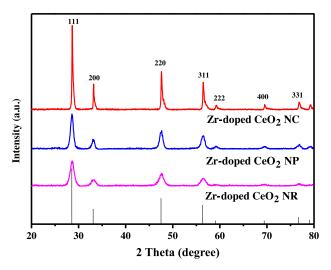
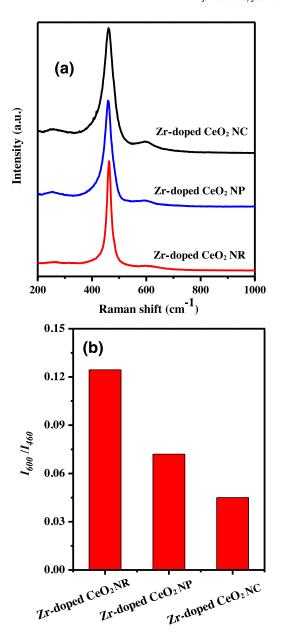


Fig. 2. XRD patterns of as-prepared Zr-doped CeO<sub>2</sub> of different morphologies.

indexed to the cubic fluorite structure of CeO<sub>2</sub> crystals (JCPDS no. 34-0394). The peaks indexed to zirconium oxide are not detected in all the samples, indicating that Zr was successfully incorporated into the ceria lattice structure. As shown in Fig. 2, the nanocubes exhibit the sharpest peaks with the highest intensity, indicating their well-developed crystallinity and large grain size, in agreement with the TEM results. Besides that, the calculated lattice parameter values for pure CeO<sub>2</sub> [23], Zr-doped CeO<sub>2</sub> nanorods, Zr-doped CeO<sub>2</sub> nanoparticles, Zr-doped CeO<sub>2</sub> nanocubes are 5.448 Å, 5.395 Å, 5.394 Å, and 5.391 Å, respectively. We can see that the lattice parameter (a) of Zr-doped CeO<sub>2</sub> of different morphologies decreased compared to pure CeO<sub>2</sub>, and this decrease is mainly due to the replacement of Ce<sup>4+</sup> (0.97 Å) by slightly smaller radius of Zr<sup>4+</sup> (0.84 Å). These XRD results further confirm the incorporation of Zr into the ceria lattice to form fluorite-like solid solutions.

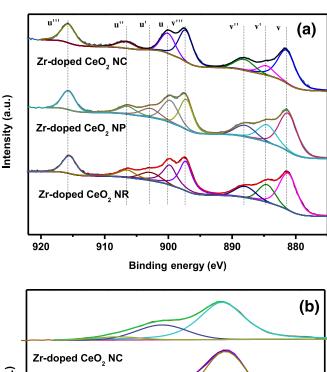
Raman spectroscopy is considered to be a very efficient and non-destructive technique for characterizations of oxygen vacancies. The Raman spectra of Zr-doped  $CeO_2$  of different morphologies are compared in Fig. 3(a). A strong Raman shift at  $\sim$ 460 nm and a relatively weak shift at  $\sim$ 600 nm can be detected. The shift at  $\sim$ 460 nm is assigned to distinct  $F_{2g}$  symmetry mode of the  $CeO_2$  phase [27,28]. Besides the band at  $460 \text{ cm}^{-1}$ , the band at  $600 \text{ cm}^{-1}$  is assigned to the non-degenerate longitudinal optical (LO) mode [29], which is related to the oxygen vacancies caused by the  $Ce^{3+}$  ion in the lattice and the small size effects [30]. According to the previous work, small cerium ions show  $Ce^{3+}$  tervalence, and in order to maintain the particles in an electrically neutral state, the lattice oxygen would escape from the structure and finally result in the formation of intrinsic oxygen vacancies, which are very



**Fig. 3.** (a) Raman spectra of Zr-doped CeO<sub>2</sub> of different morphologies, (b) the corresponding peak intensity ratio of  $I_{600}/I_{460}$ .

important for the excitation of the oxygen in substrates in the catalytic reactions. Importantly, the  $I_{600}/I_{460}$  values (Fig. 3b), which indicate the defect concentration, such as concentration of oxygen vacancies, reveal a difference among Zr-doped CeO $_2$  of different morphologies. The  $I_{600}/I_{460}$  value of Zr-doped CeO $_2$  nanorods calculated by the Raman spectra established to be the highest, and Zr-doped CeO $_2$  nanoparticles and nanocubes followed in sequence, which indicated that the concentration of oxygen vacancies of Zr-doped CeO $_2$  nanorods reached the highest if compared to nanoparticles and nanocubes.

The Ce 3d and O 1s XPS spectra of the Zr-doped CeO<sub>2</sub> nanorods, Zr-doped CeO<sub>2</sub> nanoparticles, and Zr-doped CeO<sub>2</sub> nanocubes are shown in Fig. 4, and the relative element content of Zr-doped CeO<sub>2</sub> determined are shown on Table 1. The Ce 3d spectra in Fig. 4(a) have been fitted into eight peaks:  $\nu$  ( $\sim$ 881.4 eV),  $\nu'$  ( $\sim$ 884.7 eV),  $\nu''$  ( $\sim$ 888.1 eV),  $\nu'''$  ( $\sim$ 897.3 eV),  $\mu$  ( $\sim$ 900.0 eV),  $\mu'$  ( $\sim$ 902.8 eV),  $\mu''$  ( $\sim$ 906.6 eV) and  $\mu'''$  ( $\sim$ 915.7 eV). The four  $\mu$  bands represent the Ce  $3d^{3/2}$ , and the other four  $\nu$  bands represent the Ce  $3d^{10}4$  state of Ce<sup>4+</sup> species are labeled as  $\mu$ ,  $\mu''$ ,  $\mu'''$ ,  $\nu$ ,  $\nu''$ ,



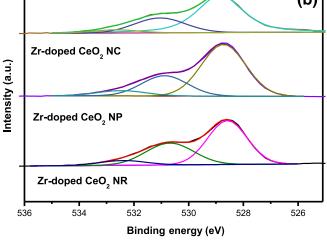


Fig. 4. (a) Ce 3d and (b) O 1s XPS spectra of Zr-doped  $CeO_2$  of different morphologies.

and  $\nu'''$ , whereas the  $3d^{10}4f^1$  state of  $Ce^{3+}$  are labeled as  $\mu'$  and  $\nu'$  [32]. From Fig 4(a), it is obvious that the  $Ce^{+4}$  oxidation state is the main chemical valence of cerium on the surface, co-existing a small amount of  $Ce^{+3}$  oxidation state which was introduced by the spontaneous transformation of  $Ce^{4+}$  (0.97 Å) to the larger  $Ce^{3+}$  (1.10 Å) [33]. Oxygen vacancies can be produced by the transformation between  $Ce^{3+}$  and  $Ce^{4+}$  as below,

$$\begin{array}{l} 4 \text{Ce}^{4+} + 0^{2^-} \rightarrow 4 \text{Ce}^{4+} + 2 e^- / \Box + 0.50_2 \rightarrow 2 \text{Ce}^{4+} + 2 \text{Ce}^{3+} \\ + \Box + 0.50_2 \end{array}$$

(□ represents an empty position)

The percentages of  $Ce^{3+}$  to the total Ce are 19.5%, 16.4% and 8.5% for the catalysts of Zr-doped  $CeO_2$  nanorods, nanoparticles and nanocubes, respectively. The higher the  $Ce^{3+}$  concentration of the total Ce that exists, the more oxygen vacancies that form. Therefore, the concentration of the oxygen vacancies of Zr-doped  $CeO_2$  nanorods reaches the maximum value.

The O 1s bands of Zr-doped  $CeO_2$  of different morphologies are shown in Fig. 4(b). The sub-bands at lower binding energy ( $\sim$ 528.5 eV) corresponded to the lattice oxygen ( $O_L$ ); ( $\sim$ 530.7 eV) corresponded to the weakly bonded oxygen species and/or chemisorbed oxygen ( $O_C$ ); the highest binding energy ( $\sim$ 532.6 eV) corresponded to the surface absorbed oxygen ( $O_S$ ), such as  $O_2^{2-}$ , or  $O^-$  belonging to defect oxide or

**Table 1.** The surface area and atomic ratios of the different catalysts.

Catalyst	$Ce^{3+}/(Ce^{4+}+Ce^{3+})^a$	Zr/(Zr+Ce)a	Zr/(Zr+Ce)b	$S_{\rm BET}~({\rm m}^2/{\rm g})$	$O_{\alpha}/O_{T}$	Crystal planes
Zr-doped CeO <sub>2</sub> NR	19.1%	11.8%	8.9%	103.9	42.3%	(110), (100)
Zr-doped CeO <sub>2</sub> NP Zr-doped CeO <sub>2</sub> NC	16.4% 8.5%	13.8% 25.8%	8.4% 8.7%	117.5 6.5	34.4% 29.6%	(111) (100), (111)

a Determined by XPS,

hydroxyl-like group [34,35]. As shown in Table 1, the ratio of  $O_{\alpha}$  to  $O_{T}$  ( $O_{T} = O_{L} + O_{C} + O_{S}$ ,  $O_{\alpha} = O_{C} + O_{S}$ ) for all samples was calculated. It is obvious that the morphology of Zr-doped  $CeO_{2}$  can effectively influence the active oxygen sites ( $O_{\alpha}$ ) on the surface. The ratio of the active oxygen sites on the surface are 42.3%, 34.4%, and 29.6% for nanorods, nanoparticles and nanocubes, respectively. At the same time, it is obvious that the concentration of lattice oxygen ( $O_{L}$ ) is diverse among the different morphologies (nanorods: 57.7%; nanoparticles: 65.6%; nanocubes: 70.4%) and the consummation of lattice oxygen indicates the formation of oxygen vacancies which are very important in the catalytic reactions.

#### 3.2. Catalytic oxidation of styrene

# 3.2.1. Catalytic activity of the Zr-doped $CeO_2$ of different morphologies

The morphology effect of the catalytic activity of Zr-doped CeO<sub>2</sub> was firstly evaluated in the oxidation of styrene to styrene oxide using 70% TBHP as the oxidant and acetonitrile as the solvent. The catalytic activities of Zr-doped CeO<sub>2</sub> of different morphologies are summarized in Table 2. The styrene conversion is 5.4% when the reaction is carried out without any catalysts, according to the autooxidation of the styrene by THBP. Also, pure CeO<sub>2</sub> or ZrO<sub>2</sub> shows a low styrene conversion about 32%. Among the Zr-doped CeO<sub>2</sub> of different morphologies, the Zr-doped CeO<sub>2</sub> nanorods exhibit the highest activity with 78.1% styrene conversion and 79.3% selectivity for styrene oxide with the highest TOF (7.656  $h^{-1}$ ) as well, followed by nanoparticles and nanocubes as shown in Table S1. This finding can be ascribed to the crystal plane effect, the higher percentage of Ce<sup>3+</sup> species and the oxygen vacancies concentration which were supported by the above analysis results. The main exposed crystal planes of Zr-doped CeO<sub>2</sub> nanorods is (110) while more stable crystal planes (111) is dominated for Zr-doped CeO2 nanoparticles. From the previous research, the stability order of the three different crystal planes is established to be (111)>(100)>(110) [36], and the more unstable the crystal plane is, the more oxygen vacancies it can produce [37]. Oxygen vacancies, which play a key role in the catalytic reactions, could activate the oxygen species

Table 2. Catalytic performance of styrene epoxidation over different catalysts<sup>a</sup>.

Catalysts	Styrene conversion (%)	Selectivity of styrene oxide (%)	Yield (%)
Zr-doped CeO <sub>2</sub> NR	78.1	79.3	61.9
Zr-doped CeO2 NP	69.5	81.1	56.3
Zr-doped CeO2 NC	44.6	88.5	39.5
$ZrO_2$	32.7	82.6	27.0
CeO <sub>2</sub>	32.2	83.1	26.8
ZrO <sub>2</sub> +CeO <sub>2</sub> <sup>b</sup>	32.2	82.2	26.5
Zr-doped CeO2 <sup>c</sup>	55.5	80.3	44.6
Blank <sup>d</sup>	5.4	81.0	4.4

 $<sup>^</sup>a$  Reaction conditions: 50 mg catalyst, 4.35 mmol styrene, 8.7 mmol TBHP (70% aqueous solution), solvent (CH $_3$ CN) = 5 mL, reaction temperature 80 °C, reaction time 12 h.

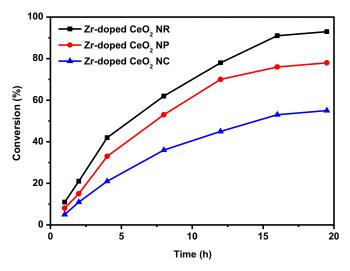


Fig. 5. Time-dependence evolution of styrene conversion for Zr-doped  ${\rm CeO_2}$  of different morphologies.

<sup>a</sup>Reaction conditions: 50 mg catalyst, 4.35 mmol styrene, 8.7 mmol TBHP (70% aqueous solution), solvent (CH<sub>3</sub>CN) = 5 mL, reaction temperature 80 °C.

in the reaction system to accelerate the catalytic process. At the same time, besides oxygen vacancies, the coordination ability of Ce-cation can influence the catalytic performance of Ce-based catalysts and it is established that Ce-cation on the (110) crystal planes has a better ability of coordination [38]. Thus, for the Zrdoped CeO<sub>2</sub> of different morphologies, the concentration of oxygen vacancies and the percentages of Ce<sup>3+</sup> to the total Ce species are the highest for nanorods, which is consistent with the reaction activity order. The catalytic activity of Zr-doped CeO<sub>2</sub> nanocubes is lower than that of nanoparticles, while the dominated exposed crystal plane of nanocubes is (100) which is the more activated. This activity difference may be due to the fact that the size of nanocubes is too big and the surface area is too small, leading to the lower absolute concentration of oxygen vacancies and less activated sites on the nanocubes catalyst.

The effect of reaction time on the conversion of styrene for Zrdoped CeO<sub>2</sub> of different morphologies is also described in Fig. 5. For nanorods, the conversion of styrene is 21.0% in the first 2 h and then increased to 78.1% with the reaction time prolonging to 12 h, while the nanoparticles and nanocubes show the similar trend during the reaction time.

Fig. 6 shows the Arrhenius plots based on the reaction rate data between  $50\,^{\circ}\text{C}$  and  $80\,^{\circ}\text{C}$ . The apparent activation energy is determined from these plots based on Eq. (1) and given in Table 3.

$$K = A \exp\left(-E_{\rm a}/RT\right) \tag{1}$$

**Table 3.** Apparent activation energy comparison of the Zr-doped CeO<sub>2</sub> of different morphologies.

Catalysts	E <sub>a</sub> (kJ/mol)		
Zr-doped CeO <sub>2</sub> NC	93.4		
Zr-doped CeO <sub>2</sub> NP	73.3		
Zr-doped CeO <sub>2</sub> NR	56.3		

b Determined by ICP

 $<sup>^{\</sup>rm b}$  The  ${\rm ZrO_2+CeO_2}$  (mol ratio of Ce/Zr = 9) was fabricated by a physical mixing method.

<sup>&</sup>lt;sup>c</sup> The Zr-doped CeO<sub>2</sub> was prepared by traditional precipitate process, with no specific morphology.

d Without catalyst.

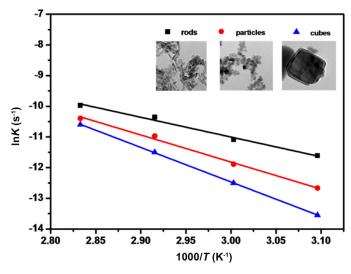


Fig. 6. Apparent activation energy comparison of the Zr-doped  ${\rm CeO_2}$  of different morphologies.

where A is the pre-exponential factor,  $E_a$  is the apparent activation energy (kJ/mol), R is the gas constant (8.3145 J/mol/K) and T is the temperature (K).

For the Zr-doped  $CeO_2$  nanorods, the apparent activation energy is 56.3 kJ/mol, which is much lower than the values of nanoparticles and nanocubes (73.3 and 93.4 kJ/mol). Therefore, the catalytic activity of Zr-doped  $CeO_2$  nanorods was higher than that of nanoparticles and nanocubes catalysts.

On the basis of the earlier reports [[39],42] and our experiment results, the possible catalytic mechanism of the Zr-doped CeO<sub>2</sub> of different morphologies is proposed. In brief, the Ce<sup>3+</sup> ions coupled with the large sized oxygen vacancy clusters are surface active sites, promoting the adsorption of styrene on the surface of catalysts. The styrene is first absorbed on the surface of Zr-doped CeO<sub>2</sub> nanomaterials through the  $\pi$ -electron interactions. Then the oxygen vacancies activate THBP in the reaction system to activated oxygen species which can react with the styrene absorbed to produce styrene oxide and leave oxygen vacancies. The oxygen vacancies are then replenished by the TBHP, thus completing the reaction cycle. Obviously, among the different morphologies, the nanorods with the highest ratio of Ce<sup>3+</sup>/(Ce<sup>3+</sup>+Ce<sup>4+</sup>) show the best catalytic performance due to the highest concentration of Ce<sup>3+</sup> and oxygen vacancies.

# 3.2.2. Catalyst reusability

The practical reusability of the Zr-doped  $CeO_2$  nanorods in the epoxidation of styrene was tested to evaluate the stability of the catalyst on behalf of Zr-doped  $CeO_2$  of different morphologies. The recycling test was performed for continuous five reaction cycles under the identical reaction as mentioned above. From Fig. 7(a), the Zr-doped  $CeO_2$  nanorods catalyst showed good stability for the epoxidation reaction in successive five runs. The stability was further confirmed by XRD patterns (Fig. 7b) of the Zr-doped  $CeO_2$  nanorods before and after five reaction recycles.

# 4. Conclusions

In summary, the Zr-doped CeO<sub>2</sub> of different morphologies, including nanorods, nanoparticles and nanocubes, were synthesized via a facile simple hydrothermal method. According to a series of characterization analysis, the Zr-doped CeO<sub>2</sub> of different morphologies possessed different surface structures, which dramatically influence the catalytic performance in the epoxidation of styrene. A correlation between different crystal morphology of Zr-doped CeO<sub>2</sub>

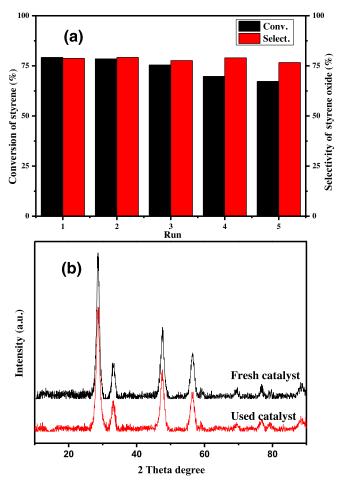


Fig. 7. (a) Recyclability test of Zr-doped CeO<sub>2</sub> nanorods for epoxidation of styrene, (b) XRD patterns of Zr-doped CeO<sub>2</sub> nanorods before and after used and TEM image of nanorods after used.

 $^aReaction$  conditions: 50 mg catalyst, 4.35 mmol styrene, 8.7 mmol TBHP (70% aqueous solution), solvent (CH $_3$ CN) = 5 mL, reaction temperature 80 °C, reaction time 12 h.

on the structure and catalytic performance was established. The Zrdoped CeO2 nanorods were found to possess the best catalytic performance for the selective oxidation of styrene to styrene oxide, followed by nanoparticles and nanocubes. On the Zr-doped CeO<sub>2</sub> nanorods, the apparent activation energy is 56.3 kJ/mol, which is much lower than the values of nanoparticles and nanocubes (73.3 and 93.4 kJ/mol). The HRTEM results indicated that (100) and (110) crystal planes are predominantly exposed for Zr-doped CeO<sub>2</sub> nanorods. The results indicated that the large sized oxygen vacancy clusters on (111) plane, and high percentage of Ce<sup>3+</sup> associated with them can efficiently improve the catalytic efficiency of styrene epoxidation. This work suggests that the catalyst could be designed to be a suitable morphology to predominately expose active planes to improve the catalytic performance. At the same time, the hydrothermal synthesis method is of easy control and environmental, which is helpful for its further industrial applications.

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#### Supplementary materials

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